

C5.2 Elasticity and Plasticity

Lecture 11 — Plasticity

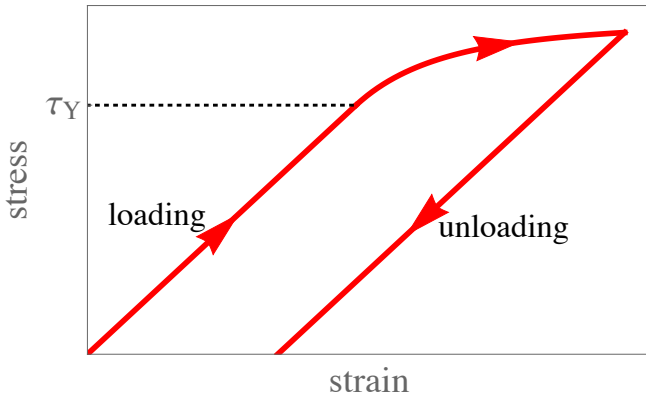
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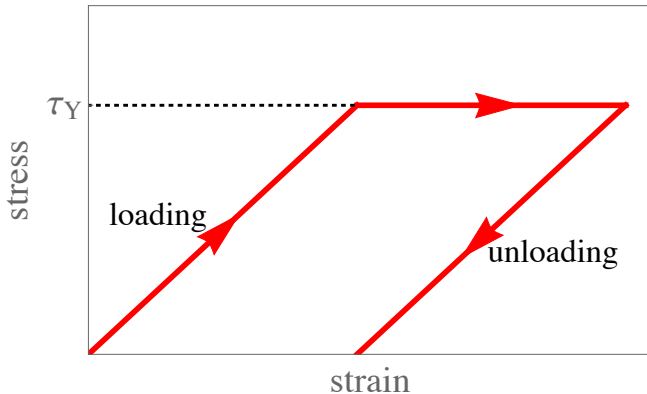
Plasticity

- ▶ **Plasticity** is commonly observed in metals.
- ▶ At sufficiently small stress, material responds elastically.
- ▶ But above a critical **yield stress** τ_Y material suffers irreversible **plastic** deformation.



Perfect plasticity

- ▶ **Perfect plasticity** is an idealised model of such behaviour.
- ▶ Material is always in one of two states:
 - ▶ **linear elastic** when below yield stress
 - ▶ **plastic** when **at** the yield stress
 - ▶ i.e. **yield stress can never be exceeded.**



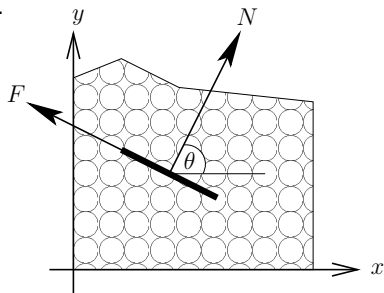
Granular plasticity

- ▶ We can construct a perfect plasticity model of a granular medium based on **Coulomb's law of friction**.
- ▶ Consider a 2D granular medium.
- ▶ Stress on interior surface element with normal and tangent vectors

$$\mathbf{n} = \begin{pmatrix} \cos \theta \\ \sin \theta \end{pmatrix} \quad \mathbf{t} = \begin{pmatrix} -\sin \theta \\ \cos \theta \end{pmatrix}$$

is given by...

$$\boldsymbol{\tau} \mathbf{n} = \begin{pmatrix} \tau_{xx} & \tau_{xy} \\ \tau_{xy} & \tau_{yy} \end{pmatrix} \begin{pmatrix} \cos \theta \\ \sin \theta \end{pmatrix} = \begin{pmatrix} \tau_{xx} \cos \theta + \tau_{xy} \sin \theta \\ \tau_{xy} \cos \theta + \tau_{yy} \sin \theta \end{pmatrix}$$



- ▶ Decompose into **normal stress** N and **tangential stress** F ...

Granular plasticity

$$\text{Normal stress } N = \mathbf{n} \cdot (\mathcal{T}\mathbf{n}) = \begin{pmatrix} \cos \theta \\ \sin \theta \end{pmatrix} \cdot \begin{pmatrix} \tau_{xx} \cos \theta + \tau_{xy} \sin \theta \\ \tau_{xy} \cos \theta + \tau_{yy} \sin \theta \end{pmatrix}$$

$$\text{i.e. } N = \frac{1}{2}(\tau_{xx} + \tau_{yy}) + \frac{1}{2}(\tau_{xx} - \tau_{yy}) \cos(2\theta) + \tau_{xy} \sin(2\theta)$$

$$\text{Tangential stress } F = \mathbf{t} \cdot (\mathcal{T}\mathbf{n}) = \begin{pmatrix} -\sin \theta \\ \cos \theta \end{pmatrix} \cdot \begin{pmatrix} \tau_{xx} \cos \theta + \tau_{xy} \sin \theta \\ \tau_{xy} \cos \theta + \tau_{yy} \sin \theta \end{pmatrix}$$

$$\text{i.e. } F = \frac{1}{2}(\tau_{yy} - \tau_{xx}) \sin(2\theta) + \tau_{xy} \cos(2\theta)$$

Granular plasticity

$$N = \frac{1}{2}(\tau_{xx} + \tau_{yy}) + \frac{1}{2}(\tau_{xx} - \tau_{yy}) \cos(2\theta) + \tau_{xy} \sin(2\theta)$$

$$F = \frac{1}{2}(\tau_{yy} - \tau_{xx}) \sin(2\theta) + \tau_{xy} \cos(2\theta)$$

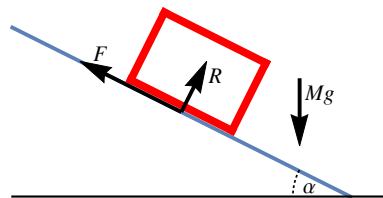
Impose constraints:

- ▶ Granular medium can't withstand any **tensile** stress, so $N \leq 0 \quad \forall \theta$ ($\Leftrightarrow \mathcal{T}$ is **negative semi-definite**)
- ▶ **Coulomb's law of friction**: $|F| \leq |N| \tan \phi \quad \forall \theta$ where ϕ is angle of friction

Aside — Coulomb's law of friction

Consider a block in frictional contact with a plane at an angle α to the horizontal

- ▶ Normal reaction R and friction force F satisfy $F \leq \mu R$ where $\mu =$ coefficient of friction.



- ▶ $F < \mu R \Rightarrow$ block sticks
- ▶ $F = \mu R \Rightarrow$ block slips

- ▶ Angle of friction $\phi = \tan^{-1} \mu$ = critical value of angle α for slip to occur.

Granular plasticity

$$N = \frac{1}{2}(\tau_{xx} + \tau_{yy}) + \frac{1}{2}(\tau_{xx} - \tau_{yy}) \cos(2\theta) + \tau_{xy} \sin(2\theta)$$

$$F = \frac{1}{2}(\tau_{yy} - \tau_{xx}) \sin(2\theta) + \tau_{xy} \cos(2\theta)$$

Impose constraints:

- ▶ Granular medium can't withstand any **tensile** stress, so $N \leq 0 \quad \forall \theta$ ($\Leftrightarrow \mathcal{T}$ is **negative semi-definite**)
- ▶ **Coulomb's law of friction**: $|F| \leq |N| \tan \phi \quad \forall \theta$ where ϕ is angle of friction
 - ▶ $|F| < |N| \tan \phi \quad \forall \theta \Rightarrow$ **material remains solid**
 - ▶ $|F| = |N| \tan \phi$ for some $\theta \Rightarrow$ material can **flow** on a **slip surface** defined by θ .

Granular plasticity

$$N = \frac{1}{2}(\tau_{xx} + \tau_{yy}) + \frac{1}{2}(\tau_{xx} - \tau_{yy}) \cos(2\theta) + \tau_{xy} \sin(2\theta)$$

$$F = \frac{1}{2}(\tau_{yy} - \tau_{xx}) \sin(2\theta) + \tau_{xy} \cos(2\theta)$$

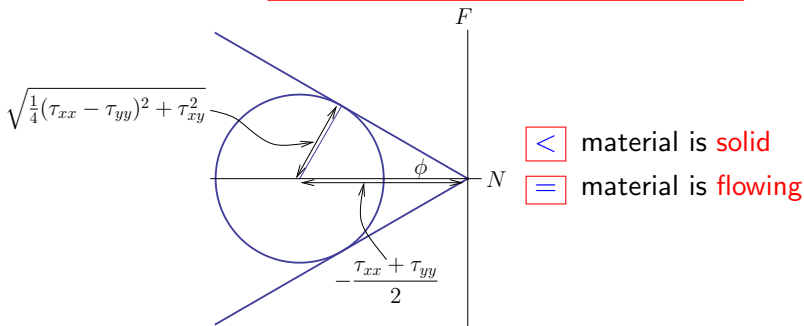
As θ varies, (N, F) lie on the **Mohr circle**

$$F^2 + \left(N - \frac{1}{2}(\tau_{xx} + \tau_{yy})\right)^2 = \frac{1}{4}(\tau_{xx} - \tau_{yy})^2 + \tau_{xy}^2$$

Granular plasticity

$$F^2 + \left(N - \frac{1}{2}(\tau_{xx} + \tau_{yy})\right)^2 = \frac{1}{4}(\tau_{xx} - \tau_{yy})^2 + \tau_{xy}^2$$

Coulomb's law \Leftrightarrow $-(\tau_{xx} + \tau_{yy}) \cos \phi \leq 2\sqrt{\tau_{xx}\tau_{yy} - \tau_{xy}^2}$



NB **yield condition** depends only on **invariants** $\text{Tr}(\mathcal{T})$ and $\text{det}(\mathcal{T})$

Granular plasticity

- ▶ For simplicity assume inertia and gravity are negligible.
- ▶ Stress components satisfy

$$\frac{\partial \tau_{xx}}{\partial x} + \frac{\partial \tau_{xy}}{\partial y} = 0$$

$$\frac{\partial \tau_{xy}}{\partial x} + \frac{\partial \tau_{yy}}{\partial y} = 0$$

- ▶ We can introduce **Airy stress function** $\mathfrak{A}(x, y)$ such that

$$\tau_{xx} = \frac{\partial^2 \mathfrak{A}}{\partial y^2} \quad \tau_{xy} = -\frac{\partial^2 \mathfrak{A}}{\partial x \partial y} \quad \tau_{yy} = \frac{\partial^2 \mathfrak{A}}{\partial x^2}$$

- ▶ While material is solid we also have elastic constitutive relations.
 - ▶ Compatibility condition $\Rightarrow \nabla^4 \mathfrak{A} = 0$.

Granular plasticity

Perfect plasticity model

Either:

- ▶ Material is **solid**

$$(\nabla^2 \mathfrak{A})^2 \cos^2 \phi < 4 \left[\frac{\partial^2 \mathfrak{A}}{\partial x^2} \frac{\partial^2 \mathfrak{A}}{\partial y^2} - \left(\frac{\partial^2 \mathfrak{A}}{\partial x \partial y} \right)^2 \right]$$

and \mathfrak{A} satisfies **biharmonic equation**

$$\nabla^4 \mathfrak{A} = 0$$

or

- ▶ Material is **flowing** (plastic)

Then \mathfrak{A} satisfies nonlinear PDE

$$(\nabla^2 \mathfrak{A})^2 \cos^2 \phi = 4 \left[\frac{\partial^2 \mathfrak{A}}{\partial x^2} \frac{\partial^2 \mathfrak{A}}{\partial y^2} - \left(\frac{\partial^2 \mathfrak{A}}{\partial x \partial y} \right)^2 \right]$$

Granular plasticity

elastic $\nabla^4 \mathfrak{A} = 0$ (E)

plastic $(\nabla^2 \mathfrak{A})^2 \cos^2 \phi = 4 \left[\frac{\partial^2 \mathfrak{A}}{\partial x^2} \frac{\partial^2 \mathfrak{A}}{\partial y^2} - \left(\frac{\partial^2 \mathfrak{A}}{\partial x \partial y} \right)^2 \right]$ (P)

- ▶ In general we have to solve a **free boundary problem**:
 - ▶ (E) in solid regions
 - ▶ (P) in flowing regionswith the boundary between them unknown *a priori*.
- ▶ (E) is elliptic, but (P) is **hyperbolic**
(cf Monge–Ampère equation)
 - ▶ transports information along characteristics corresponding to slip surfaces.

Granular plasticity

elastic $\nabla^4 \mathfrak{A} = 0$ (E)

plastic $(\nabla^2 \mathfrak{A})^2 \cos^2 \phi = 4 \left[\frac{\partial^2 \mathfrak{A}}{\partial x^2} \frac{\partial^2 \mathfrak{A}}{\partial y^2} - \left(\frac{\partial^2 \mathfrak{A}}{\partial x \partial y} \right)^2 \right]$ (P)

- ▶ In principle solution of (E), (P) and suitable BCs determines \mathfrak{A} and therefore stress components $\tau_{xx}, \tau_{xy}, \tau_{yy}$
- ▶ but no way to find **velocity** (\dot{u}, \dot{v}) when material is flowing.
- ▶ If inertia was important then we would be stuck...

$$\rho \frac{\partial \dot{u}}{\partial t} = \frac{\partial \tau_{xx}}{\partial x} + \frac{\partial \tau_{xy}}{\partial y} \qquad \rho \frac{\partial \dot{v}}{\partial t} = \frac{\partial \tau_{xy}}{\partial x} + \frac{\partial \tau_{yy}}{\partial y}$$

- ▶ Situation is even worse in 3D (3 components of momentum equation + 1 yield condition for 6 stress components τ_{ij})
- ▶ Need a **flow rule** to close the problem...

Dislocations

Example: consider antiplane strain $\mathbf{u} = w(x, y)\mathbf{e}_z$ with $\nabla^2 w = 0$

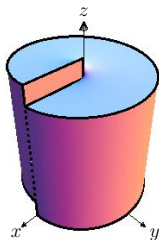
- ▶ Note that (plane polars (r, θ))

$$w(x, y) = \frac{b}{2\pi} \tan^{-1} \left(\frac{y}{x} \right) = \frac{b}{2\pi} \theta$$

satisfies Laplace's equation for $(x, y) \neq (0, 0)$.

- ▶ If $0 \leq \theta < 2\pi$ then w jumps by b across positive x -axis.
- ▶ This displacement corresponds to a **cut-and-weld** operation...

Dislocations



- 1 **Cut** cylinder along +ve x -axis
- 2 Shift one side up a distance b
- 3 **Weld** the two faces back together

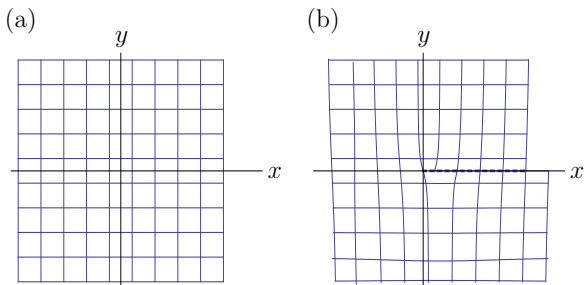
- ▶ Once everything is welded back together, the cylinder is in a state of **self-stress**, with

$$\tau_{\theta z} = \frac{\mu b}{2\pi r}$$

- ▶ (The z -axis is a source of **incompatibility**.)

Dislocations

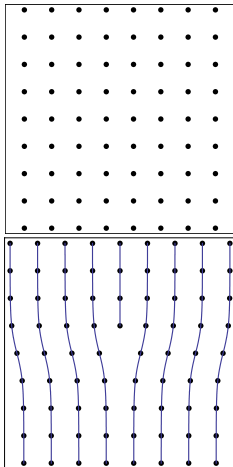
We can perform a similar cut-and-weld operation in plane strain. . .



- ▶ Again cut along positive x -axis.
- ▶ Now shift one side **horizontally** with respect to the other before welding back together.
- ▶ What happens if we do this operation at an **atomic** level?

Dislocations

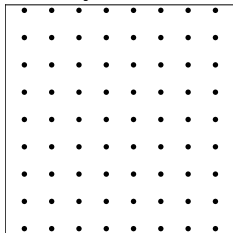
- ▶ Start with a pristine lattice.
- ▶ Cut along positive x -axis.
- ▶ Shift lower side **by an atomic distance** and reconnect.
- ▶ It's like inserting an extra column of atoms into the lattice.
- ▶ This configuration is called an **edge dislocation**.
- ▶ Corresponding antiplane strain configuration is a **screw dislocation**



Dislocations

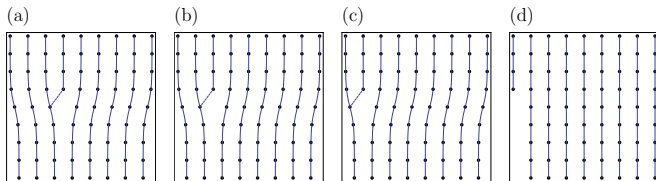
Atoms in a metal are usually arranged on a periodic crystal lattice.

- ▶ If lattice was **pristine** then the **yield stress** required to cause irreversible deformation would be of order μ .
- ▶ Measured yield stresses are **much** smaller (by factor of $\approx 10^{-5}$!)

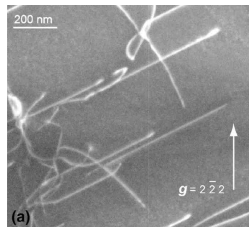


- ▶ **Hypothesis** (1930s): plastic deformation arises from motion of **dislocations** through the lattice. . .

Dislocations



- ▶ Only small reorganisation of lattice is needed to shift dislocation and cause irreversible deformation of the lattice.
- ▶ Requires yield stress τ_Y much smaller than μ .
- ▶ Hypothesis was confirmed **much** later by electron microscopy: metals contain trillions of dislocations which are generated and propagated by plastic deformation of the sample.



Perfect plasticity in metals

Study of behaviour of dislocations leads to the following...

Hypothesis: plastic deformation in metals is driven by **shear stress**
— normal stress does not (usually) cause irreversible deformation

Based on this hypothesis, build a **perfect plasticity** model
whereby...

- ▶ $|\text{shear stress}| < \tau_Y \Rightarrow$ **elastic**
- ▶ $|\text{shear stress}| = \tau_Y \Rightarrow$ **plastic**